# Designing slurry conditions to control size distribution of spray-dried dense YSZ granules 

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#### Abstract

This study suggests approaches to achieve the desired size and size distribution of highly dense spherical granules by investigating the effect of slurry conditions on size distribution. Highly dense spherical granules were prepared with a solid content of over $77 \mathrm{wt} \%$ by spray-drying the slurry. A prolonged deagglomeration time of 64 h provided adequate flowing ability by breaking up almost all aggregates and improved dispersibility, resulting in reduced granule sizes and narrow size distributions. The optimum slurry conditions for maintaining dispersibility were $1 \mathrm{wt} \%$ of the dispersant and a strong basic pH , which had the greatest effect on size distribution. Based on these considerations, the $10.6 \mu \mathrm{~m}$ sized $3 \mathrm{~mol} \%$ yttria-stabilized zirconia granules were synthesized with $99.83 \%$ density, $97.17 \%$ sphericity, and uniform size with fraction yield of $80.01 \%$ at $10-20 \mu \mathrm{~m}$. These dense granules have significantly higher hardness and modulus values of 19.19 and 206.68 GPa , respectively, than that of pellet and film types. To the best of our knowledge, the relationship between the slurries and the span of the size distribution of ceramic granules during spray-drying has been demonstrated for the first time.


## KEYWORDS

mechanical properties, particle size distribution, slurry optimizing, spray-drying, yttriastabilized zirconia

## 1 | INTRODUCTION

Yttria-stabilized zirconia (YSZ) is widely used as a standard material in thermal barrier coatings because of its low thermal conductivity, ${ }^{1}$ dental restoration owing to its excellent mechanical strength, ${ }^{2}$ and applications, such as grinding and milling media, requiring high density with uniform size and sphericity. ${ }^{3}$ Recently, the demand for ultrasmall (less than $30 \mu \mathrm{~m}$ ) beads, which are smaller in size than existing ceramic beads, has been rapidly increasing
for the nano-miniaturization of core raw material powders and integration of parts in various high-tech industries, such as electricity/electronics, energy, and environment. ${ }^{4,5}$ In particular, nano-miniaturization and high-purity raw material powders are necessary for the high performance of multilayer ceramic capacitors. Therefore, the size of the required beads has been gradually reduced, and a higher hardness is required than before to withstand strong impact. There is a technical limitation in manufacturing ultrasmall ceramic beads using the existing ceramic bead
manufacturing process. In addition to simply making the bead size small, it should be possible to impart high hardness and high-density characteristics to the ceramic beads themselves so as to impart to the material to be pulverized high purity and miniaturization. It is also important to achieve precise size control and high sphericity. Therefore, research on controlling the size of spherical YSZ to a small size with high mechanical strength and density is important. Furthermore, because the method of classifying and supplying ceramic beads with a wide particle size has to be price competitive, developing a processing method for ceramic particles that satisfy the required characteristics in a single size is necessary.

Controlled wet processing is an important step in ceramic manufacturing and significantly impacts the morphologies and properties of all intermediate and final products in ceramic manufacturing, including spraydrying. ${ }^{6-15}$ Granulation is an essential step in improving the flowability of submicron and nano-ceramic powders to make them suitable for industrial processing. ${ }^{16,17}$ Controlling the average granule size and size distribution is important in granulation processes. Spray-drying is a method of fabricating ceramic granules by which a water-based suspension (slurry) is changed to a dry, free-flowing spherical powder (granules) by rapidly spraying the feeding slurry into a stream of heated gas for drying. ${ }^{18-20}$ The main parameters affecting the shape and size distribution of spray-drying technology are slurry properties (concentration, viscosity, stability, and properties of initial particles) and process parameters (atomizing pressure of spraying gas, temperature, feed rate, and nozzle diameter). ${ }^{21-24}$

Therefore, the effects of slurry conditions on the properties of spray-dried ceramic granules were studied. Most studies have focused on the effect of slurries on the morphology and final product properties of granules. ${ }^{9,25-27}$ For example, concerning the spray-drying of alumina, a correlation was established between the slurry characteristics and the morphologies of the granules; a flocculated slurry is prone to producing smaller and dimpled granules. ${ }^{28}$ In addition, the condensing degree of the granules can be affected by the moisture content and stability of the slurry, indicating that the density of the green body can be changed by the slurry, resulting in the slurry properties indirectly controlling the structure and properties of the sintered body. ${ }^{12}$ Coddet et al. established a stability map showing stable regions for controlling the morphology of zirconia granules based on the amount of dispersant and binder depending on the sedimentation height using a single droplet drying device. ${ }^{29-31}$

Few researchers have focused on the technology for producing $10 \mu \mathrm{~m}$ class granules via spray-drying. Most studies have produced an average granule size of over $50 \mu \mathrm{~m}$ and exhibited a wide size distribution ( $\sim 50 \mu \mathrm{~m}$ or greater
spread) (Table S1). Our research group succeeded in fabricating highly dense YSZ ceramic granules of size $20-30 \mu \mathrm{~m}$ by controlling the structural features of the supraparticle, such as sphericity and micromechanical properties, by adjusting the spray-drying process parameters. ${ }^{7}$ However, the yield of the desired particle size was approximately $20 \%$ because only the process parameters of spray-drying were applied, without considering the particle size distribution. Previous studies have attempted to understand the effect of spray-drying process variables and the relationship between the properties of the slurry and the morphology of the granules. The direct relationship between the properties of the slurry and the size, including the distribution of the granules, has yet to be adequately studied because the spraying process parameter has a more significant effect on the size change. However, to make $10 \mu \mathrm{~m}$ class granules with uniform size, it is imperative to not only control the spraying process parameters but also finely optimize the slurry conditions. Therefore, acquiring highly dense ultrasmall YSZ granules with a spherical morphology and a narrow size distribution while achieving mechanical strength is crucial, yet challenging. It is particularly difficult to achieve a uniform granule size with a breakthrough yield of the desired size.

Herein, we report the highly dense $10 \mu \mathrm{~m}$ size of $3 \mathrm{~mol} \%$ YSZ granules with a high yield fraction of over $80 \%$ by controlling the slurry conditions. To the best of our knowledge, this is the first study to demonstrate the relationship between slurries and the size distribution of ceramic granules during spray-drying. We also examined the modification of the shape of the granules. Here, we present four approaches to these objectives: (i) preparing high-concentration YSZ-based aqueous suspensions considering the viscosity, density, spherical morphology, and size distribution of sintered granules; (ii) investigating the influences of deagglomeration time on the flowing ability of slurry and the size distribution of granules, and the effect of atomizing pressure on the size change; (iii) studying the dispersing ability of the slurry according to the amount of dispersant and pH to determine suitable stabilization conditions; and (iv) acquiring $10 \mu \mathrm{~m}$ class YSZ granules with superior sphericity, narrow size distribution, and excellent mechanical strength by optimizing the spray-drying conditions.

## 2 | EXPERIMENTAL PROCEDURES

## 2.1 | Materials

In this study, zirconia powder containing $3 \mathrm{~mol} \% \mathrm{yttria}$ was used (average particle size: $4.68 \mu \mathrm{~m}$, supplied by CENOTEC, Korea). High-purity $\alpha$-alumina powder
(AKP-30, Sumitomo, Japan) was added within the solubility limit to prevent abnormal grain growth and act as a sintering aid. ${ }^{32}$ To achieve well-stabilized and well-dispersed zirconia slurries, an ammonium polycarboxylate dispersant (San Nopco, SN-DISPERSANT 5468) was added. The pH was adjusted using diluted solutions of hydrochloric acid (HCl, 1 N, Samchun Chemicals, Korea) or sodium hydroxide solution ( $\mathrm{NaOH}, 1 \mathrm{~N}$, Samchun Chemicals, Korea).

## 2.2 | Slurry preparation

The slurry was prepared with various conditions, including solid contents (65.25-80.25 wt\%), time of ball-mill treatment ( $1-64 \mathrm{~h}$ ), dispersant amount ( $0.5-2 \mathrm{wt} \%$ ), and pH (5.5-10.5), to investigate the stable and appropriate conditions for $10 \mu \mathrm{~m}$ class granule size. The slurry was composed of $3 \mathrm{~mol} \% \mathrm{YSZ}$ (65-80 wt\%), $\mathrm{Al}_{2} \mathrm{O}_{3}$ ( $0.25 \mathrm{wt} \%$ ), dispersant ( $0.5-2 \mathrm{wt} \%$ ), and distilled water ( $17.75-32.75 \mathrm{wt} \%$ ), so that the ratio of the ceramic powder/slurry was $65.25-$ $80.25 \mathrm{wt} \%$. The mixture was ball-milled using 2 mm balls that were $5 / 3$ the weight of the slurry in a polyethylene bottle under controlled pH conditions.

## 2.3 | Spray-drying and sintering process

Spray-drying of the suspensions was carried out in a twofluid nozzle spray-dryer (SD 1010, EYELA, Japan), and the setting parameters were established by referring to our previous experiments. ${ }^{7}$ The inlet and outlet stream temperatures were maintained at 110 and $70-80^{\circ} \mathrm{C}$, respectively. Furthermore, the feed rate and drying airflow were fixed at $600 \mathrm{~mL} / \mathrm{h}$ and $0.64 \mathrm{~m}^{3} / \mathrm{min}$, respectively. The conditions of atomizing pressure ( $100-250 \mathrm{kPa} \mathrm{N}_{2}$ gas pressure) were changed to investigate the effect of reducing the size of the granules. The sprayed and dried powders were sieved with mesh sizes of $10,20,25,32,45$, and $75 \mu \mathrm{~m}$ to determine the fraction yield efficiency. The sieved powder was sintered in an alumina crucible at $1400^{\circ} \mathrm{C}$ for 80 min at a heating rate of $2^{\circ} \mathrm{C} / \mathrm{min}$.

## 2.4 | Characterizations

The microstructures of the YSZ granules were analyzed using field-emission scanning electron microscopy (SEM, IT300, JEOL, Japan). Granule size distributions were determined by static light scattering using a particle size analyzer (LS320, Beckman Coulter Inc., USA). Granule density was measured using a gas pycnometer (BELpycno, MicrotracBEL Corp., Japan). The viscosity changes of the
slurries at 100 rpm were characterized using a viscometer (DV-II, Brookfield, USA) with spindle No. 64, and the zeta potential was determined using dynamic light scattering (Malvern ZS Nano S analyzer, Malvern Instrument Inc., London). In addition, the stability of each sample was analyzed by measuring the backscattering (BS) every hour in a Turbiscan Lab apparatus (Formulaction, France) at $25^{\circ} \mathrm{C}$. Turbiscan carried out step-by-step vertical scanning of the whole sample using a pulsed near-infrared light source ( $\lambda=850 \mathrm{~nm}$ ) and converted the macroscopic aspects of the mixtures into graphics. Backscatter data acquisition from the backscatter detector received light backscattered by the sample at $135^{\circ}$ from the incident beam. ${ }^{33}$

Mechanical testing was performed using nanoindentation. Before the nanoindentation tests, the granules were embedded in a black epoxy resin (Struers, DuroFast, Denmark). The molten resin was then hot pressed ( $F=10 \mathrm{kN}$ and $T=180^{\circ} \mathrm{C}$ for 10 min ). The surfaces of the granules were polished to a roughness of $2 \mu \mathrm{~m}$ using a sequence of ultrafine 1000, 2000, and 4000 grit SiC papers. The surface roughness of the mounted specimens was analyzed using a 3D measuring laser microscope (Olympus OLS5000 LEXT). The mechanical tests were performed in an Anton Paar nanoindenter (NHT3, Anton Paar, Austria) system with a load of $5-20 \mathrm{mN}$ using a diamond Berkovich tip (angle of $142.3^{\circ}$ and tip roundness of $r \leq 100 \mathrm{~nm}$ ) indenter. The loading and unloading rates were both 10.0 Hz , and the holding time at the maximum load was 3 s . The average hardness and elastic modulus values were determined from 10 indentation experiments.

## 3 | RESULTS AND DISCUSSION

## 3.1 | The effect of solid contents

Spray-drying was performed with solid contents of 65.25 , $75.25,77.25$, and $80.25 \mathrm{wt} \%$. Figure 1 shows the YSZ granules spray-dried at different solid contents. The drying temperature was controlled by adjusting the inlet temperature of the spray-drying process, which appropriately controlled the drying kinetics. For the formation of highdensity ceramic granules, the evaporation velocity of the water solvent was optimized to be slower than the diffusion rate during mass transfer, generating highly dense granules by the isotropic shrinkage of liquid droplets. Therefore, spray-drying was carried out at inlet and outlet stream temperatures of 110 and $70-80^{\circ} \mathrm{C}$, respectively. Meanwhile, owing to the high moisture content at a solid content of 65.25 and $75.25 \mathrm{wt} \%$, less-dried granules easily merge during the drying process; thus, the morphology presents poor sphericity.


FIGURE 1 Scanning electron microscopy (SEM) micrographs of sintered granules with various solid contents of (A) $65.25 \mathrm{wt} \%$, (B) $75.25 \mathrm{wt} \%$, (C) $77.25 \mathrm{wt} \%$, and (D) $80.25 \mathrm{wt} \%$ after sintering.


FIGURE 2 (A) $D_{10}, D_{50}, D_{90}$, and span values of sintered granules; (B) viscosity of slurry and density of sintered granules with various solid contents.

The concentration of the slurry, which depends considerably on the solid content, is an important parameter that determines the size of the dried granules. In Figure 2A, as the solid contents increase from 65.25 to $80.25 \mathrm{wt} \%$; there is a tendency for $D_{10}, D_{50}$, and $D_{90}$ to increase from $3.91,8.25$, and 12.43 to $14.97,29.51$, and $57.84 \mu \mathrm{~m}$, respectively. To define the distribution width, the span of the volume-based size distribution is defined as

$$
\begin{equation*}
\text { Span value }=\frac{D_{90}-D_{10}}{D_{50}} \tag{1}
\end{equation*}
$$

A higher value indicates a wider particle size distribution. Owing to the relatively small increase in $D_{10}$ compared to $D_{50}$ and $D_{90}$, the span value increased with the increase in solid content up to $77 \mathrm{wt} \%$. In particular, at the point of change from 77.25 to $80.25 \mathrm{wt} \%, D_{10}, D_{50}$, and $D_{90}$ increase significantly overall, and the span value remains constant. This result reveals that the solid contents had a significant effect on the particle size and size distribution and shows that solid contents of $80.25 \mathrm{wt} \%$ are not suitable for $10 \mu \mathrm{~m}$ granules.

The higher concentration of solids in the feeding slurry resulted in higher viscosity, as shown in Figure 2B.


FIGURE 3 (A) Low solid contents; (B) High solid contents; (C) High \& Agglomerated solid contents. Shematic illustration of the effect of the conditions of slurry contents on the size and density of the dried granule with a slower evaporation velocity of water solvent than diffusion rate.

Consequently, when a feeding slurry with a high solid content is atomized, large droplets dry into large granules. This is because they shrink less when the liquid evaporates compared to droplets produced from a feeding slurry with low solid content, as shown in Figure 3A,B. In contrast, at low solid contents, there were smaller droplets and more shrinkage, resulting in a smaller granule size. This relationship can also be expressed by the following equations that describe the droplet size and geometric diameter of the dry particles ${ }^{18,20}$ :

$$
\begin{gather*}
D_{d}=\alpha\left(\frac{\sigma^{s} \rho_{L}^{t 1} \eta^{r}}{\rho_{A}^{t 2}}\right)\left(\frac{f_{L}}{f_{L} v_{L}+f_{A} v_{A}}\right)  \tag{2}\\
D_{g}=\sqrt[3]{\frac{C_{F}}{\rho_{\mathbf{P}}}} D_{d} \tag{3}
\end{gather*}
$$

where $D_{d}$ and $D_{g}$ are the average diameter of the droplet and geometric diameter of the dried granule after the spray-drying process, respectively. $\alpha$ is a constant that depends on the spray design, $\rho$ is the density, $\eta$ is the viscosity, $\sigma$ is the surface tension, $f$ is the flow rate, and $v$ is the velocity of the liquid $(L)$ and the atomization gas $(A) . s$, $t$, and $\gamma$ are constants, the values of which depend on operating conditions. $C_{F}$ is the concentration of the feed slurry, and $\rho_{P}$ is the droplet particle density. The above formulas indicate that the particle size is also directly affected by changes in concentration, varying with the solid content of the feed solution. As the solid content increased, $D_{d}$ increased because the viscosity and density of the liquid
increased, whereas the other variables remained constant. The geometric diameter also increased with an increase in $D_{d}$ and feed slurry concentration.

As shown in Figure 2B, all the solid contents have a high density of over $96 \%$ of the relative density (theoretical density: $\left.6.04 \mathrm{~g} / \mathrm{cm}^{3}\right)^{2}$ due to the relatively high solid contents of over $65.25 \mathrm{wt} \%$. Among the conditions, the density of the sintered granules showed the highest value at a solid content of $77.25-80.25 \mathrm{wt} \%$, along with good sphericity. To optimize the conditions for the $10 \mu \mathrm{~m}$ class of YSZ granules with relatively superior yield, sphericity, and mechanical properties, the following research was conducted with $77.25 \mathrm{wt} \%$ solid contents.

## 3.2 | Effect of deagglomeration time

Figure 4 shows the viscosity of the slurry, the density of sintered granules, size distribution after sintering, and morphology of dried granules to investigate the influence of the degree of deagglomeration on the flow behavior of a slurry containing $77.25 \mathrm{wt} \%$ solid content according to the ball-milling time. As shown in Figure 4A, the initial viscosity at 1 h was relatively low. In the initial ball-milling stage of less than 1 h , the powder seemed to agglomerate into large clumps, resulting in the formation of large oval granules approximately $50 \mu \mathrm{~m}$ in size after spray-drying (Figure 4D). Subsequently, the viscosity increased from 1 to 4 h , likely because the size of the powder was significantly reduced from the initial size owing to a weak connection between the initial particle clumps and decreased


FIGURE 4 (A) Viscosity of slurry and density of sintered granules with various solid contents; (B) size distributions of granules obtained by different deagglomeration times; (C) $D_{10}, D_{50}, D_{90}$, and span values of the sintered granules. Scanning electron microscopy (SEM) micrographs of spray-dried granules with a deagglomeration time of (D) 1, (E) 4, (F) 24, (G) 40, and (H) 64 h .
polydispersity of the particles. A smaller particle size results in a higher number of particles in a given volume, and as a result, the particle surface area can be orders of magnitude higher than the initial one, leading to greater interactions between particles and increased viscosity. Additionally, the size distribution of the particles is narrower, which may result in less efficient packing compared to particles with a larger distribution. It means that less free space is available in the case of narrow distribution for individual particles to move around, making sample flow less easily. Furthermore, the time is not enough for the complete spreading of the dispersant in this position. As a result, during the initial period when the powder size and size distribution reduced rapidly, the suspension's viscosity can be increased. This speculation can be further confirmed by the change in the size distribution of the granules after sintering, as shown in Figure 4B. The granules spray-dried after deagglomerating for 1 h showed one primary population of fine particles centered at approximately $11 \mu \mathrm{~m}$ and a secondary, smaller population of coarse particles centered at approximately $50 \mu \mathrm{~m}$ owing to the initial conglomerated particles. As the deagglomeration time increased, the population of coarse particles diminished, and the size of the fine particles was reduced, suggesting that the clumped raw powders were destroyed. Therefore, as shown in Figure 4C, the span value decreases until the coarse particles disappear after 24 h .

In addition, extending the deagglomeration time from 4 to 64 h significantly reduced the viscosity, suggesting a distinctive improvement in the fluidity of the suspension. Specifically, part of the liquid trapped inside the network structure is released, a more ordered structure is formed, and the dispersant is distributed evenly and well. Meanwhile, the particle size and size distribution of the
spray-dried granules did not change significantly as the deagglomeration time increased from 24 to 64 h , as shown in Figure 4C, because the agglomerates of the initial particles were almost completely broken after 4 h . The change in viscosity due to the reduction in particle size and size distribution is insignificant, likely because the change in particle size is small. However, it appears that the dispersant is uniformly dispersed, and the dispersibility seems to improve with increasing ball mill time. Consequently, a uniform structure was formed in the slurry, and its flowing ability further improved, resulting in fine granules, as shown in Figure 4F-H. This size distribution result follows Equation (2), which shows the proportionality between viscosity and droplet size.

The density displays an opposite trend to that of the viscosity, as shown in Figure 4A. It is believed that the large clumps of the initial particles are strongly agglomerated during drying so that the spray-dried granules show a high density of $97.8 \%$ at 1 h . At a deagglomeration time of 4 h , the density decreased because of the agglomerated solid content and low dispersibility and fluidity of the suspension, which occurred when the initial large clumps began to break up, as shown in Figure 3C. At the 24 h stage, almost all agglomerates of the initial powder disappeared, and the well-dispersed slurry with an ordered structure had denser particle packing within the granules compared to the initial condition (Figure 3B). As the time increased up to 64 h , the density reached $99.8 \%\left(6.03 \mathrm{~g} / \mathrm{cm}^{3}\right)$.

## $3.3 \mid$ Effect of atomizing pressure

Atomizing pressure is one of the major variables governing granule size in spray-dry process settings. The


FIG URE 5 Scanning electron microscopy (SEM) micrographs of spray-dried granules after sintering at each atomizing pressure: (A) 100 , (B) 200 , and (C) 250 kPa ; (D) $D_{10}, D_{50}, D_{90}$, and span values of the sintered granules at each atomizing pressure.
average droplet size was inversely proportional to the atomizing energy. ${ }^{18}$ Figure 5 shows the morphology and size distribution of the granules at atomizing pressures of 100,200 , and 250 kPa . When gas collides with the surface of the liquid at a high atomizing pressure, the liquid accelerates and shrinks. Conversely, the atomization energy was reduced at lower pressures, resulting in larger droplets, and the final dried granules were larger in size. This finding is consistent with the literature on spray-dried hydroxyapatite ${ }^{34,35}$ and metal powders. ${ }^{25,36}$ The span value, which depends on the atomizing pressure, is consistent because of the size change at a constant rate, as shown in Figure 5D. In addition, the effect of the nozzle diameter was checked, but there was no effect on the size change of the granules acquired from the cyclone chamber, depending on the nozzle diameters of 0.71 and 0.41 mm (Figure S 1 ). For forming $10 \mu \mathrm{~m}$ sized granules, the highest atomizing pressure of 250 kPa allowed in the equipment is more suitable. However, for the formation of uniform granules with a low span value of approximately $10 \mu \mathrm{~m}$, the slurry conditions, such as stability, should be modified, as shown in the following results.

## 3.4 | Influence of dispersant amount and $\mathbf{p H}$ on the stability of slurry

As shown in Figure 3B,C, for small-sized granules, the deagglomeration of the particles in the slurry should be maintained; the stability of the slurry is extremely important for ultrasmall granules. To determine the suitable amount of dispersing agent, the zeta potential and viscosity


FIGURE 6 Effect of dispersant amounts on the zeta potential and viscosity of slurry.
of the slurry were measured ranging from 0.5 to $2 \mathrm{wt} \%$, as shown in Figure 6. The value of the zeta potential indicates the possible behavior of the dispersion, and a high value is related to suspension stability. At the isoelectric point (IEP) of the powder at which the electrokinetic surface charge of the particle is zero, the attractive interaction between particles is not countered, and the suspension achieves its most flocculated state. Consequently, it is expected that the surface charge of YSZ in the region far from the IEP value is sufficient to induce a repulsive force between the particles to maintain the slurry in a dispersed state. The amount of dispersant that provides optimum potential stability with the lowest zeta potential has the lowest viscosity at a


FIGURE 7 Scanning electron microscopy (SEM) micrographs of sintered granules after sintering with (A) 1 and (B) $2 \mathrm{wt} \%$ dispersant amount; (C) size distributions and (D) fraction yield of granules on each sieve opening size after sieving obtained by 1 and $2 \mathrm{wt} \%$ of dispersant amount.
dispersant amount of $1 \mathrm{wt} \%$. It appears to have a low viscosity because the powder is spread with strong repulsion at the part with the lowest zeta potential. This shows that there is a correlation between the viscosity and amount of dispersant. Thus, it can be inferred that $1 \mathrm{wt} \%$ dispersant is the optimum amount to disperse and stabilize the slurry. Determining the optimal dispersant amount is important for improving the potential stability of the slurry and enhancing the efficiency of the deagglomeration process.

Figure 7 presents the results of the SEM micrograph, size distribution, and fraction yield after classifying the sintered granules using different sieve sizes, with 1 and $2 \mathrm{wt} \%$ of dispersant amounts while all other experimental conditions remain constant. For the optimized dispersant amount of $1 \mathrm{wt} \%$, clear differences were detected in the size distribution with a narrow graph after sintering compared with $2 \mathrm{wt} \%$, as shown in Figure 7C. Additionally, the yield for $10-20 \mu \mathrm{~m}$ during the sieving process with dry granules significantly increased from $23.35 \%$ with a $2 \mathrm{wt} \%$ dispersant amount to $67.65 \%$ with a $1 \mathrm{wt} \%$ dispersant amount, in Figure 7D.

To determine the optimal pH for the slurry with $1 \mathrm{wt} \%$ dispersant, the effect of pH on the zeta potential was investigated, as shown in Figure 8A. Because of the weakly acidic dispersant and raw material powder, the slurry prepared without pH adjustment had a pH of 6.5 .

Usually, the zeta potential farthest from the IEP with a stable pH zone is found in the alkaline part. ${ }^{5,23}$ This is because the anionic polyelectrolyte dispersant at an acidic pH significantly reduces the degree of dissociation of the deflocculant, thus reducing its adsorption efficiency. As expected, the zeta potential was over 30 mV at a pH of 8 or more, and a well-dispersed slurry was obtained. This observation is further proven by the Turbiscan Lab Expert stability analyzer investigating the stability of the slurry at pH 6.5, 8.5, and 10.5, as shown in Figure 8B-D. Turbiscan is a potentially promising method for studying stability over a short period while simultaneously providing information on destabilization kinetics, including purification, coalescence, agglomeration, and sedimentation. ${ }^{33,37}$ Based on the variation in the droplet volume fraction (migration) or average size (coalescence), variation of the BS signals measured as a function of time was performed along the cylindrical cell every 10 min for 12 h . At pH 6.5 , there is a significant change in delta BS, which forms at a height slightly above the floor due to rapid sedimentation and flocculation. However, only slight sedimentation occurred at pH 8.5. Because of the improved dispersion stability of the slurry, the thickness of the sedimentation layer also decreased as the pH increased to 10.5 . From this result, the pH of the stabilized slurry was determined to be 10.5.



FIGURE 8 (A) Effect of pH on zeta potential of slurry: (B-D) the Turbiscan profiles showing delta backscattering as a function of time (0:00-12:00 h) at pH (B) 6.5 , (C) 8.5 , and (D) 10.5 , respectively.

## 3.5 | The characterization and mechanical strength of $\mathbf{1 0} \boldsymbol{\mu m}$ YSZ granules

Figure 9A,B shows the morphologies of the granules after sintering and sieving with a $10-20 \mu \mathrm{~m}$ mesh at optimized slurry conditions (solid contents of $77.25 \mathrm{wt} \%$, deagglomeration time of $64 \mathrm{~h}, 1 \mathrm{wt} \%$ dispersant amount, and pH 10.5 ), and atomizing conditions at 250 kPa based on the above experimental results. Figure S2 shows the X-ray diffraction results of the YSZ raw powder and YSZ granules before and after sintering, where the monoclinic phase was transformed to the tetragonal phase after sintering. The average diameter of the granules was $10.6 \mu \mathrm{~m}$ with a uniform size, varying by only $1.84 \mu \mathrm{~m}$ using ImageJ software. Sphericity was calculated by measuring the radius of 100 granules, using the following equation:

$$
\begin{equation*}
\text { Sphericity }(\%)=\frac{d_{s}}{d_{L}} \times 100 \tag{4}
\end{equation*}
$$

where $d_{s}$ is the ratio of the maximum axial length, and $d_{L}$ is the minimum axial length of a particle. The calculated sphericity showed an excellent value of $97.17 \%{ }^{38}$ In addition, the $D_{50}$ of the size distribution after sieving with a 10-20 $\mu \mathrm{m}$ mesh had similar results with an average of $10.18 \mu \mathrm{~m}$, as shown in Figure 9C. Under these conditions, the fraction yield after sieving at $10-20 \mu \mathrm{~m}$ was $80.01 \%$, a $40 \%$ increase over the conditions before controlling the pH (Figure 7D, 67.65\%). Figure 10 shows the effects of slurry
stability on granule size and size distribution. The large size distribution was significantly reduced compared with the results before the dispersant amount and pH conditions were adjusted. A better dispersed slurry could be used to fabricate smaller granules of uniform size. The span value was significantly reduced by optimizing the slurry conditions.

Consequently, the desired $10 \mu \mathrm{~m}$ YSZ granules with superior sphericity and extremely high yield efficiency were synthesized by carefully considering the slurry preparation parameters and spray-drying setting parameters. The mechanical properties of the fabricated YSZ granules were analyzed by nanoindentation, as shown in Figure 11. In nanoindentation, a diamond tip is pressed into the specimen to given maximum depth or load and then removed. Simultaneously, the load and displacement of the indenter were also recorded. From such a load-depth curve, the hardness $(H)$ and Young's modulus $(E)$ of the specimen can be analyzed using the Oliver and Pharr method. ${ }^{39}$ The Poisson ratio of YSZ is $0.25,{ }^{40}$ which is needed to determine Young's modulus. For an accurate measurement, the surface of the polished specimens was checked using the 3D surface roughness analysis technique, and the value was less than $2 \mu \mathrm{~m}$ (Figure 11A).

At different nanoindenter loads, the hardness and elastic modulus were analyzed to investigate the optimal conditions for the nanoindentation test. The highest hardness is observed at 10 mN . When the load was increased to 20 mN , the elastic modulus increased, but the hardness decreased;


FIGURE 9 (A and B) Scanning electron microscopy (SEM) micrographs of sintered granules with the optimized condition for $10 \mu \mathrm{~m}$ class granules; (C) size distributions before and after sieving 10-20 $\mu \mathrm{m}$, and (D) fraction yield of granules on each sieve opening size after sieving.


FIGURE $10 \quad D_{10}, D_{50}, D_{90}$, and span values of sintered granules with $2 \mathrm{wt} \%$ dispersant amount-no pH control, $1 \mathrm{wt} \%$ dispersant amount-no pH control, and $1 \mathrm{wt} \%$ dispersant amount- pH 10.5 .
therefore, the measurement was conducted at 10 mN . The YSZ granules have excellent hardness and elastic modulus of 19.19 and 206.68 GPa , respectively. The hardness and elastic modulus of the epoxy polymer was only 3.63 and 25.78 GPa , respectively. A previous nanoindentation examination of YSZ pellets ${ }^{40}$ and coated film ${ }^{41}$ showed hardness and Young's modulus of 24.7 and 257 GPa , and of 7.23 and

122 GPa , respectively. In previous research of YSZ granules, the hardness and Young's modulus were 26.77 and 210.19 GPa with $20 \mu \mathrm{~m}$ sized granules ${ }^{7}$; our results are not significantly lower despite being more influenced by the polymer due to a high surface area. This result is a relatively high value, even though the size is smaller and not in film or pellet form; therefore, it can be greatly affected by the polymer wrapping the granule around it.

The details of all the experimental conditions are listed in Table 1. These results have remarkable industrial relevance for applications requiring high-density granulation, such as grinding and milling media. Ultrasmall size of $10 \mu \mathrm{~m}$ YSZ granules with superior sphericity and yield efficiency were synthesized by optimizing the conditions of the slurry. The mechanical strength was sufficiently high for these applications. Moreover, this result presents a new direct approach to spray-drying by optimizing the slurries to control not only the size but also the size distribution.

## 4 | CONCLUSIONS

The purpose of this study was to determine the influence of slurry conditions on the granule size and size distribution of highly dense YSZ granules synthesized using spray-drying. Furthermore, granules of size $10 \mu \mathrm{~m}$ were synthesized by adjusting the conditions of the slurry with


FIGURE 11 (A) 3D surface roughness of the polished surface, (B) hardness and elastic modulus at each load, and (C) load-displacement curve of mounted and polished yttria-stabilized zirconia (YSZ) granules after sintering at the optimized conditions for $10 \mu \mathrm{~m}$ granules and that of epoxy. The inset is an optical image of the measured YSZ granules.

TABLE 1 Summary of the experimental conditions and analysis.

| Conditions |  |  |  |  |
| :--- | :--- | :--- | :--- | :--- | :--- | :--- |
| Solid |  |  |  |  |
| contents |  |  |  |  |
| $(\mathbf{w t \%})$ |  |  |  |  |

the solid contents to form dense granules in deagglomeration, a stable state that maintains high fluidity, and controlling the spray-drying parameters, including the atomizing pressure.

As the solid content of YSZ increased, the size of the granules increased because of the large droplets resulting from the increase in viscosity and reduced shrinkage during evaporation. The size distribution
and density increased but became constant as the solid content increased. A longer deagglomeration time was effective in reducing the size distribution and granule size, but the effect became increasingly insignificant. At the starting point, the viscosity increased with the breaking of the initial conglomeration of the clumped particles to a uniform size; but the longer the time, the viscosity decreased with more improved flowing ability. The density was improved by enhancing the deagglomeration and dispersion as the deagglomeration time increased. Increasing the atomizing pressure reduced the size of the droplet, thus reducing the size of the final granule; however, the size distribution did not change significantly. The stability of the slurry had a significant effect on the size distribution; therefore, it should be adjusted with the proper amount of dispersant and pH . In this study, the degree of stabilization was determined through zeta potential, viscosity, and Turbiscan analyses. The obtained dense YSZ granules showed an average size of $10.6 \mu \mathrm{~m}$, a high sphericity of $97.17 \%$, significantly improved narrow size distribution with a high fraction yield of $80.01 \%$, and excellent mechanical properties with a hardness of 19.19 GPa and an elastic modulus of 206.68 GPa.

This study shows that the size and distribution of $10 \mu \mathrm{~m}$ granules can be finely adjusted by controlling the conditions of the slurry with a breakthrough in the yield. The yield, which was $20 \%$ in our previous study, has dramatically increased to $80.01 \%$. It is expected that the key technology of this study can be applied not only to grinding and milling media but also to various applications requiring granulation.

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## CONFLICT OF INTEREST STATEMENT

The authors declare that they have no known conflict of interests or personal relationships that could have influenced the work reported in this study.

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## SUPPORTING INFORMATION

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